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### ENERGY ABSORPTION BEHAVIOUR OF SHORT FIBRE COMPOSITE

A.K.M. Masud A.K.M. Kais Bin Zaman

#### ABSTRACT

Plastic composite material is one of the most suitable design materials in aircraft industries, due to its high specific strength. As unidirectional composites exhibit anisotropy, much effort has been drawn by the researchers to make isotropic plastic composite materials. One of their efforts is to make randomly oriented short fibre composites. Unfortunately, theoretical interpretation is highly case dependent. Energy absorption is an important property for structural design element especially for aircraft industries. Thus in this research a generalized model is developed to calculate energy absorption by the randomly oriented short fibre composite. Here, classical single- fibre pull-out process is considered for calculating the energy of the randomly oriented short fibre composite. Concept of single- fibre pull-out test is incorporated to the model, such that the fibres intersecting a fracture plane (where the fracture of composite occurs) would pull out or fracture like single- fibre pull-out test. The energy contribution of each fibre in the fracture plane is calculated for different embedded length and orientation to estimate the total fracture energy. According to the model, it is found that the energy absorption depends upon the intrinsic properties of the fibre and matrix, volume fraction, critical embedded length of the fibre (the critical embedded length is defined as the length at which the load required to pull out the fibre becomes almost equal to the fracturing load of the fibre, during the singlefibre pull-out test.), length of the short fibre and orientation of the fibres in matrix. The significance of the explicit formula for the energy absorption is that it can be used to estimate the amount of fibres and fibre length required to absorb certain amount of energy for composite to be used in structural design element before the material is made.

#### **1. INTRODUCTION**

Plastic composites have been in use for long due to their lightweight, high specific strength and improved performance under stringent physical, chemical and environmental conditions. From a technological point of view, short fibre composites are well established because they are easy to fabricate.

Department. of Industrial and Production Engineering, Bangladesh University of Engineering and Technology ,Dhaka – 1000, Bangladesh

Prediction of the mechanical properties of the composites is a complex problem, due to many factors that play a considerable role in the properties of the composites. The energy absorbed by the composite depends upon the intrinsic properties of the fibre and the matrix and their interface. Fibre and matrix work together to develop internal energy to resist the external applied load. The load applied to the composite is transmitted along fibre and matrix as well as their interface. The classical single- fibre pull out test is a common means to determine the interfacial properties of fibre and matrix. Here, the critical embedded length is defined as the length at which the load required to pull out the fibre becomes almost equal to the fracturing load of the fibre, during the single- fibre pull-out test. This critical embedded length can easily be determined with the help of single fibre pull-out test. During fracture of short fibre composite, it can be assumed that some fibres are pulled out and some would fracture at the fracture plane of the composite. Whether a fibre in the fracture plane would pull out or fracture that depends on the critical embedded length. In this model contribution of energy by each fibre and matrix itself is calculated to estimate the total fracture energy of the composite.

#### 2. MODELING OF ENERGY ABSORPTION

#### 2.1. Single-fibre pull-out process

Single- fibre pull-out test is a very easy and common process of estimating interfacial properties of fibre and matrix. Many researchers have found that load required to pull out from embeddedment increases with the embedded length until the fibre fracture load attains. Although this increment has little deviation from linearity, it is assumed to be linear for simplicity. Load response for single-fibre pull-out test is shown in Figure 1[1].



Figure. 1 Representation of load required to pull-out and fracture of fibre during pull out process.

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#### 2.2. Mathematical modeling of energy absorption

It is considered for randomly oriented short fibre composite that applied force is distributed three dimensionally in the matrix and the fibres. To resist the applied external energy, there would develop an internal energy inside the body of the composite. Both the matrix and the fibre would contribute in this development of internal energy. The sum of energy absorption by matrix and the fibre would be the total energy absorbed by the composite during fracture process. It is considered that ultimate failure would occur in an imaginary plane *S* that is perpendicular to the axis of applied load.

The equation of the energy absorption by the composite  $W_c$  is assumed to be calculated by the following equation: [1, 2]

$$W_{c} = \sum_{i=1}^{N_{f}} W_{i}^{*} + \frac{(1 - V_{f})}{(1 + AV_{f})} W_{m}$$
$$= N_{f} W_{i}^{*} + \frac{(1 - V_{f})}{(1 + AV_{f})} W_{m}$$
(1)

where,  $V_f$  is the fibre volume fraction and  $W_m$  is the fracture energy of the matrix.  $N_f$  is the number of the fibres per unit area those intersect plane S,  $W_i^*$  is the energy contributed by the individual fibre on the fracture plane,  $\overline{W}_i^*$  is the expectation of the energy  $W_i^*$ , A is the shape parameter [2] which is related to the packing condition of the fibre. First part of Eq. (2.1) is the contribution of fibre and the second part defines the contribution of matrix in the composite.

From statistical metallography [3], it is found that in case of randomly oriented short fibre composite, the number of fibre intersecting any plane in a unit volume is:

$$N_f = (N_v I_f) / 2$$

where,  $N_v$  and  $I_f$  are the number and length of the short fibres in a unit volume, respectively.

$$N_{f} = \frac{N_{v} l_{f}}{2} = \frac{V_{f}}{2A_{f}}$$
(2)

where,  $A_f$  is the cross-sectional area of the short fibre.

#### 2.3. Estimation of individual energy contribution by each fibre w<sub>i</sub>\*

Figure 2 shows a typical situation of forces that would occur in the composite during loading. It is assumed that a fibre AOB would intersect an imaginary plane S along which the composite is expected to fracture. F is the force that is developed along the axis of the fibre during the loading process.

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Here,  $\phi$  is the angle between loading direction and fibre axis. When plane *S* intersects the fibre at point O, it has two parts. The shorter length  $l_x$  would act as embedded length and this will determine whether the fibre would be fractured or pulled out. Maximum length of  $l_x$  would be half of the total length of the fibre [4].



Figure. 2 Condition of the force acting along the fibre axis that appears in the fracture plane.

To assign the contribution of fibre orientation angle and the embedded length in the energy absorbed by the fibre, it is assumed that the energy contributed by a single fibre  $Wi^*$  is a functional product of the function of embedded length u(x) and the angular position of the fibre  $v(\phi)$ , because these two variables are varying simultaneously.

$$W_{i}^{*} = W_{i}^{*}(x,\phi) = u(x)v(\phi)$$
<sup>(3)</sup>

It is evident from the pull-out test and its results (Figure 1) that when the embedded length is less than the critical length  $I_c$ , the fibres will be pulled out and the load required to pull out the fibre would increases linearly with the embedded length. Again it is evident from Fig. 1 that when the embedded length is greater than the critical one, the fibres will be fractured out but this fracture load does not depend further on the embedded length.

When the embedded length is less than the critical one, it is assumed that interfacial property would only contribute to the energy absorption. Thus, when the embedded length is less than the critical one, then energy contribution due to the embedded length u(x) would be calculated from Figure 1 as  $\frac{1}{2} (A_f \sigma_f / l_c) \cdot x^2$ . On the other hand, when the embedded length is greater than the critical one,

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then energy contribution due to embedded length would depend on the energy of fibre fracture and would be calculated from Figure 1 as  $\frac{1}{2} A_f \sigma_f \delta$ . Here, it is also assumed that the fibre would shear internally when the embedded length is greater than the critical one and would be calculated as  $\tau_y A_f \phi x$ . This idea of shearing the fibre itself and the calculation of energy have been taken from Ref. 3. Therefore, the energy contribution by each fibre for different embedded length is represented as follows:

$$u(x) = \begin{cases} \frac{1}{2} \frac{A_f \sigma_f}{l_c} x^2 & l_f \leq 2l_c \\ \frac{1}{2} A_f \sigma_f \delta + \tau_y A_f \phi x & l_f > 2l_c \end{cases}$$
(4)

where,  $\sigma_f$  is the strength of the fibre,  $A_f$  is the cross-sectional area of the short fibre, x is the arbitrary length of the fibre,  $\delta$  is the fracture deformation of the fibre and  $\tau_v$  is the inter-laminar shear stress of the fibre itself.

It is assumed that tangential component  $F_t$  in Figure 2 has no contribution in the energy absorption of the composite. Thus, the energy absorption by a single fibre mainly depends upon the angular position of the fibre with respect to the loading direction. Therefore, the function of orientation angle is equal to  $\cos\phi$ , i.e.

$$v(\phi) = \cos \phi \tag{5}$$

From Eq. (2.5) it is evident that among the fibres in the fracture plane *S*, those that have a small inclination would hold extra energy compared to those with large orientation angle but along the fracture plane ( $\phi = 90^{\circ}$ ), the fibre would have no contribution in the energy absorption.

The applied load would be distributed among all the fibres at the fracture plane. The whole combination of the fibres in the fracture plane would not fail until all the fibres in that fracture plane S are peeled. When all the fibres are peeled, the specimen would gain the maximum internal energy contributed by the fibres and after this point fracture of the composite would occur through out the plane S.

#### 2.4. Distribution of x and $\phi$

L

To calculate the expected value of individual energy contributed by a single fibre ,  $W_i^*$ , the probability distributions of *x* and  $\phi$  should be calculated. A hemisphere is considered with radius  $l_f$  as shown in Figure 3, where  $l_f$  is the length of the short fibre [1]. Here, all the intersecting fibres in the fracture plane *S* of the composite are assumed to be accumulated at the center of the hemisphere. Thus, these fibres would be distributed randomly in three dimensions at the center of the hemisphere.

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Figure. 3 Schematic diagram of a hemisphere where the fibres are homogeneously distributed.

To calculate  $P_f$ , the probability that the fibre would have the required embedded length  $I_f$  and it would have the required orientation in between  $\phi$  and  $\phi+d\phi$ , is assumed to be the ratio of the area of the ring element (Figure 3) made by angle  $d\phi$  to the area of the hemisphere [1].

$$P_{f} = \frac{2\pi l_{f} \times l_{f} \sin \phi \ d\phi}{\frac{1}{2} \times 4\pi \ l_{f}^{2}}$$
(6)  
$$= \sin \phi \ d\phi$$

The probability of locating the fibre that intersects plane *S* (Fig. 3) with an angle  $\phi$  is defined by  $P_s$  and can be obtained by the following equation:

$$P_{s} \propto \cos \phi$$
  

$$\Rightarrow P_{s} = K_{\phi} \cos \phi$$
(7)

where,  $K_{\phi}$  is the proportionality constant.

For total probability,

$$\int_{0}^{\pi/2} K_{\phi} \cos \phi \, d\phi = K_{\phi} = 1$$

$$\Rightarrow P_s = \cos \phi \tag{8}$$

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Further, the probability of locating the embedded length  $I_x$  in the length between x and x+dx is defined by  $P_x$  and can be obtained by the equation: where,  $K_x$  is the proportionality constant.

$$P_x \propto dx$$
$$\Rightarrow P_x = K_x \, dx \tag{9}$$

For total probability,

$$\int_{0}^{l_{f}/2} K_{x} dx = 1$$

$$\implies K_{x} = \frac{2}{l_{f}}$$
(10)

#### 2.5. Calculation of the expected value of the energy contribution:

It is mentioned earlier that the individual contribution of fibre in energy  $W_i^*$  depends on:

- (a) Embedded length of the fibre,  $I_x$
- (b) Angle of orientation  $\phi$  with respect to the loading direction
- (c)  $P_{f_{f}}$  the probability that the fibre would have the required embedded length  $I_{f}$ and would have the required orientation in between  $\phi$  and  $\phi + d\phi$
- (d)  $P_s$ , the probability that the fibre would intersect the fracture surface as well as would have a small orientation  $\phi$  with the loading direction
- (e)  $P_x$ , the probability of locating the embedded length of the fibre in the length ranges between x and x+dx.

To have the contribution of short fibre in the energy of the composite, the abovementioned factors should occur simultaneously and hence  $W_i^*$  shall be the product of all the factors from (a) to (e).The differential values for the energy contribution by the *i*-th fibre can be written as follows:

$$dW_i^* = W_i^* (x, \phi) K_x dx K_\phi \cos\phi \sin\phi d\phi$$
(11)

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Thus, the expected energy absorption in the fracture plane by a single fibre  $W_i^*$ would be the integration of  $dW_i^*$ .

$$\overline{W}_{i}^{*} = \int_{0}^{L_{f}/2} \int_{0}^{\pi/2} W_{i}^{*}(x,\phi) K_{x} dx K_{\phi} \cos\phi \sin\phi d\phi$$
$$= K_{x} K_{\phi} \int_{0}^{L_{f}/2} u(x) \int_{0}^{\pi/2} \cos\phi^{2} \sin\phi d\phi$$
$$= \frac{K_{x} K_{\phi}}{3} \int_{0}^{L_{f}/2} u(x) dx$$
(12)

Here, the embedded length is varied from 0 to  $l_f/2$ , each fibre would be divided into two parts and Ir /2 would be the maximum length that can participate in the energy absorption. Again, the orientation angle  $\phi$  is varied from 0 to  $\pi/2$  as fibres along the loading direction ( $\phi = 0^{\circ}$ ) have more contribution and along the fracture plane ( $\phi = 90^{\circ}$ ) it would have no contribution.

In case of composite, made of short fibres having a fibre length of less than 21c, all the fibres would pull out. Thus the equation of energy contributed by pull-out process by each fibre is as follows:

$$\overline{W_{i}^{*}} = \frac{K_{x} K_{\phi}}{3} \int_{0}^{l_{f}/2} \frac{1}{2} \frac{A_{f} \sigma_{f}}{l_{c}} x^{2} dx$$
$$= \frac{A_{f} \sigma_{f} l^{2}_{f}}{72 l_{c}}$$
(13)

In case of composite made of short fibres having a fibre length greater than 21c. the energy absorbed would depend upon three phenomenon. In this composite, some of the fibres intersect the plane S with embedded length less than critical length and some would be greater than the critical length and accordingly some fibres would pull out and some would fracture out. Thus, energy would absorb during pull-out and fracturing of the fibres in the fracture plane. Further we assume that certain energy would absorb during internal shearing of the fibres.

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 $\overline{W_i}^*$  = pull-out energy + fracture energy + shear energy

$$\overline{W_i^*} = K_x K_\phi \int_0^{l_c} \int_0^{\pi/2} \frac{1}{2} \cdot \frac{A_f \sigma_f}{l_c} x^2 dx \cdot \cos^2 \phi \sin \phi \, d\phi$$
$$+ K_x K_\phi \int_{l_c}^{L_f/2} \int_0^{\pi/2} \frac{1}{2} \sigma_f A_f \, \delta \, dx \cdot \cos^2 \phi \, \sin \phi \, d\phi$$
$$+ K_x K_\phi \int_{l_c}^{L_f/2} \int_0^{\pi/4} \tau_y A_f \, \phi \, x \, dx \cdot \cos^2 \phi \, \sin \phi \, d\phi \tag{14}$$

Now, from Eq. (2.14),

Pullout energy:

$$(\overline{W_i^*})' = K_x K_{\phi} \int_0^{l_c} \int_0^{\pi/2} \frac{1}{2} \cdot \frac{A_f \sigma_f}{l_c} x^2 dx \cdot \cos^2 \phi \sin \phi d\phi$$
$$= \frac{A_f \sigma_f l_c^2}{9l_f}$$
(14a)

Fracture energy:

$$(\overline{W_i^*})^{"} = K_x K_{\phi} \int_{L_c}^{l_f/2} \int_{0}^{\frac{\pi}{2}} \frac{1}{2} \sigma_f A_f \delta dx \cos^2 \phi \sin \phi d\phi$$
$$= \frac{A_f \sigma_f \delta}{3l_f} \left(\frac{l_f}{2} - l_c\right)$$
(14b)

Shear energy:

$$(\overline{W_i^*})''' = K_x K_{\phi} \int_{l_c}^{L_f / 2\pi/4} \int_{0}^{\pi/4} \sigma_y A_f \phi x \, dx \cdot \cos^2 \phi \sin \phi \, d\phi$$
$$= 0.0517 \frac{A_f \sigma_f}{l_f} (\frac{l_f^2}{4} - l_c^2) \qquad (14c)$$

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Now putting the individual contribution by the pull-out, fracture and shear from Eqs. (2.14.a), (2.14.b) and (2.14.c) in Eq. (2.14), the expected value of the energy absorbed by single fibre is written as follows:

(2.15) 
$$\overline{W_{i}^{*}} = \frac{\sigma_{f}A_{f}l_{c}^{2}}{9l_{f}} + \frac{\sigma_{f}A_{f}\delta}{3l_{f}} \left(\frac{l_{f}}{2} - l_{c}\right) + 0.0517 \frac{\sigma_{f}A_{f}}{l_{f}} \left(\frac{l_{f}^{2}}{4} - l_{c}^{2}\right)$$
$$= 0.0594 \frac{\sigma_{f}A_{f}l_{c}^{2}}{l_{f}} + \frac{\sigma_{f}A_{f}\delta}{3l_{f}} \left(\frac{l_{f}}{2} - l_{c}\right) + 0.0129 \sigma_{f}A_{f}l_{f} (15)$$

For composite reinforced with fibres of length  $I_f \leq 2I_c$ , the energy absorption of the composite is obtained by putting the value of Eq. (2.13) and Eq. (2.2) in Eq. (2.1):

(2.16) 
$$(W_c)_{l_f \le 2l_c} = \frac{\sigma_f V_f l_f^2}{144 l_c} + \frac{(1 - V_f)}{(1 + AV_f)} W_m$$
 (16)

For composite reinforced with fibres of length  $l_f > 2l_o$ , the energy absorption of the composite is obtained by putting the value of Eq. (2.15) and Eq. (2.2) in Eq. (2.1):

$$(W_c)_{l_f > 2l_c} = \frac{\sigma_f V_f}{2} [ 0.0594 \quad \frac{l_c^2}{l_f} + \frac{\delta}{3l_f} (\frac{l_f}{2} - l_c) + 0.0129 \quad l_f ]$$

$$(2.17) \qquad \qquad + \frac{(1 - V_f)}{(1 + AV_f)} W_m$$

$$(17)$$

# 3. EFFECTS OF FIBRE VOLUME FRACTION AND EMBEDDED LENGTH ON ENERGY ABSORPTION

Two sets of graphs have been plotted in Figure 4 and 5 using the Eqs. (2.16) and (2.17) and considering the data of Carbon fibre and Polyester resin. The energy absorbed by the matrix,  $W_m$ , the strength of the fibre,  $\sigma_f$  and the fracture deformation,  $\delta$  are considered as 3 kJ/m<sup>2</sup>, 2700×10<sup>3</sup> kN/ m<sup>2</sup> and 10 m $\mu$ , respectively [1,5]. Here, *A* is chosen as 2.5 as an approximate upper limit [2].

In Figure 4, the fibre length is taken as less than or equal to twice the critical embedded length. Here, graphs have been plotted for this case considering three different values of  $l_t/l_c$  ratio. When  $l_t/l_c$  is equal to 1, fibre length is equal to critical embedded length, i.e., fibre length is less than twice the critical embedded length. When  $J_t/l_c$  ratio is 1.5 or 2, fibre length is twice the critical embedded length.

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Now putting the individual contribution by the pull-out, fracture and shear from Eqs. (2.14.a), (2.14.b) and (2.14.c) in Eq. (2.14), the expected value of the energy absorbed by single fibre is written as follows:

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$$= 0.0594 \frac{\sigma_{f}A_{f}l_{c}^{2}}{l_{f}} + \frac{\sigma_{f}A_{f}\delta}{3l_{f}} \left(\frac{l_{f}}{2} - l_{c}\right) + 0.0129 \sigma_{f}A_{f}l_{f} (15)$$

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$$\left(W_{c}\right)_{l_{f} \leq 2l_{c}} = \frac{\sigma_{f} V_{f} l_{f}^{2}}{144 l_{c}} + \frac{(1 - V_{f})}{(1 + AV_{f})} W_{m}$$
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$$(W_{c})_{l_{f}>2l_{c}} = \frac{\sigma_{f}V_{f}}{2} [0.0594 \quad \frac{l_{c}^{2}}{l_{f}} + \frac{\delta}{3l_{f}}(\frac{l_{f}}{2} - l_{c}) + 0.0129 \quad l_{f}]$$

$$(17) \qquad \qquad + \frac{(1 - V_{f})}{(1 + AV_{f})} W_{m}$$

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Two sets of graphs have been plotted in Figure 4 and 5 using the Eqs. (2.16) and (2.17) and considering the data of Carbon fibre and Polyester resin. The energy absorbed by the matrix,  $W_m$ , the strength of the fibre,  $\sigma_f$  and the fracture deformation,  $\delta$  are considered as 3 kJ/m<sup>2</sup>, 2700×10<sup>3</sup> kN/m<sup>2</sup> and 10 m $\mu$ , respectively [1,5]. Here, *A* is chosen as 2.5 as an approximate upper limit [2].

In Figure 4, the fibre length is taken as less than or equal to twice the critical embedded length. Here, graphs have been plotted for this case considering three different values of  $I_t/I_c$  ratio. When  $I_t/I_c$  is equal to 1, fibre length is equal to critical embedded length, i.e., fibre length is less than twice the critical embedded length. When  $I_t/I_c$  ratio is 1.5 or 2, fibre length is twice the critical embedded length.

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(2

Here, these conditions have been plotted using different fibre length of 1 mm, 2 mm, 3 mm, 4 mm and 5 mm and shown in Fig. 4 (a~e). It may be mentioned here that all the fibres will be pulled out because the length of the fibre is less than or equal to twice the critical embedded length. From fig. 4, it is evident that energy absorption increases with the increase in fibre volume fraction, fibre embedded length and the  $I_f /I_c$  ratio, when fibre embedded length is less than or equal to critical embedded length.



Figure. 4 Effects of fibre length on energy absorption when the fibre length is less than or equal to twice the critical embedded length

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In Figure 5, the fibre length is greater than twice the critical embedded length. Here again, graphs have been plotted for this case considering three different values of If /lc ratio, equal to 2.1, 2.5 and 3. For all conditions, fibre length is greater than twice the critical embedded length. Here, these conditions have also been plotted using different fibre length of 1 mm, 2 mm, 3 mm, 4 mm and 5 mm and shown in Figure 5 (a~e). It may also be mentioned here that some of the . fibres will be pulled out and some will be fractured because the length of the fibre is greater than twice the critical embedded length. It is seen from the Figure 5 that when the  $I_f / I_c$  ratio is just greater than 2, energy absorption increases with the increase in fibre volume fraction and fibre length. But when the  $l_f / l_c$  ratio is takes as 3, composites exhibit different behavior of energy absorption, that is, energy absorption decreases compared to the energy absorbed when the  $l_f / l_c$ ratio is just greater than 2. From the graph, it is seen that, energy is also increased with the increase in volume fraction and embedded length but the amount of energy absorbed is less than that of the cases when a smaller ratio of  $l_{\ell}/l_c$  is used. The reason behind this is assumed that when  $l_{\ell}/l_c$  is greater, i.e., when fibre embedded length is much greater than critical embedded length, fracturing of fibre dominates in the fracturing process of composites compared to the pulling out of the fibres. It is assumed that the amount of pull out energy is higher than that of fibre fracture energy. As a result, energy absorption of composites shows such behavior as depicted in Figure 5.

So, from Figure 4 and Figure 5, it is evident that absorbed energy increases with the increase in fibre volume fraction and fibre length up to a limit of  $l_f / l_c$ . After that limit energy tends to decrease as fibre fracture dominates over fibre pull-out in the fracturing process of composite materials.

#### 4. CONCLUSION

In this paper, a theoretical micro-mechanics model is developed to estimate the energy absorbed by composites. An explicit expression of the energy absorption is derived as a function of volume fraction of fibre, embedded length and the properties of the fibre and the matrix. The significance of the explicit formula for the energy absorption is that it can be used to estimate the amount of fibres and fibre length required to absorb certain amount of energy for the composite before the material is made.

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